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## Ultrafast Optical Microscopy of Single Monolayer Molybdenum Disulfide Flakes

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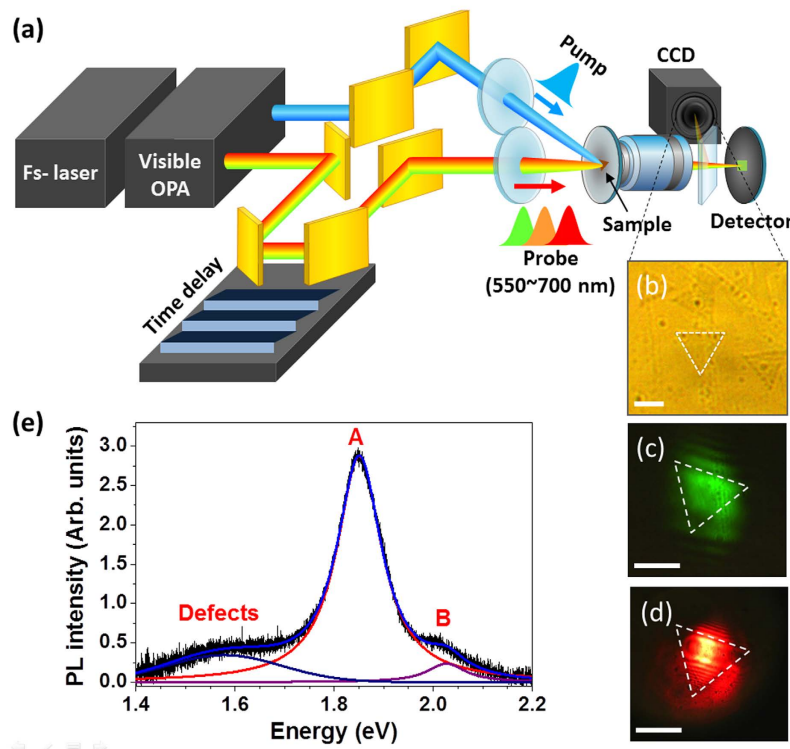
We have performed ultrafast optical microscopy on single flakes of atomically thin CVD-grown molybdenum disulfide, using non-degenerate femtosecond pump-probe spectroscopy to excite and probe carriers above and below the indirect and direct band gaps. These measurements reveal the influence of layer thickness on carrier dynamics when probing near the band gap. Furthermore, fluence-dependent measurements indicate that carrier relaxation is primarily influenced by surface-related defect and trap states after above-bandgap photoexcitation. The ability to probe femtosecond carrier dynamics in individual flakes can thus give much insight into light-matter interactions in these two-dimensional nanosystems.

There has been an explosion of recent interest in the quasi-two-dimensional (2D) transition metal dichalcogenides (TMDCs), which are layered materials with strong in-plane covalent bonding, leading to atomically thin layers within their structure<sup>1</sup>. Their unique 2D geometry allows for the design and fabrication of complicated structures at desirable positions within optoelectronic devices, much more efficiently than other low-dimensional nanomaterials<sup>1,2</sup>. In particular, molybdenum disulfide (MoS<sub>2</sub>) has received much recent attention due to its excellent electronic<sup>3</sup> and optical properties<sup>4,5</sup>. Most notably, the indirect bandgap in the bulk (~1.2 eV) changes to a direct bandgap (~1.9 eV) as the number of atomic layers is reduced to one, causing MoS<sub>2</sub> monolayers to generate strong photoluminescence<sup>4-7</sup>. Therefore, unlike graphene, which does not have an intrinsic band gap<sup>8</sup>, TMDCs are especially promising for potential optoelectronic applications, such as photo-transistors<sup>9</sup>, photodetectors<sup>10</sup>, and electroluminescent devices<sup>11</sup>. Many of these applications will depend on a detailed knowledge of the optical properties and carrier relaxation dynamics, which can be elucidated by photoexciting carriers and probing their relaxation as a function of layer thickness with femtosecond time resolution.

Here, we use ultrafast optical microscopy (UOM)<sup>12,13</sup> to measure ultrafast carrier dynamics in atomically thin single molybdenum disulfide flakes grown by chemical vapor deposition (CVD). By tuning the probe photon energy through the MoS<sub>2</sub> band gap (both indirect and direct), our UOM measurements show that conduction and valence band states are rapidly populated on a sub-picosecond (ps) time scale in a MoS<sub>2</sub> monolayer after photoexcitation at 3.1 eV, consistent with previous work<sup>14-18</sup>. Pump fluence-dependent measurements reveal that subsequent carrier relaxation in our samples is primarily due to surface-related defects and trap states, not the Auger processes observed in previous measurements on MoS<sub>2</sub> and other semiconductor nanosystems<sup>12,18-23</sup>. We also observed an increase in the carrier relaxation time with an increase in the number of MoS<sub>2</sub> layers, likely due to the well known layer-dependent changes in the electronic structure<sup>16,24</sup>. Our UOM measurements of photon energy- and fluence-dependent carrier dynamics in single MoS<sub>2</sub> flakes thus shed light on carrier relaxation in these nanosystems, which should influence future optoelectronic and nanoelectronic applications<sup>15,16,25</sup>.

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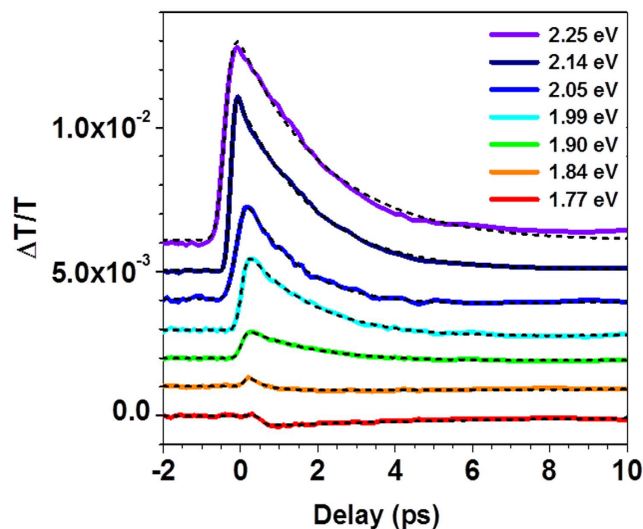


**Figure 1. Experimental setup for ultrafast optical microscopy, as well as microscopic images and photoluminescence spectra of single MoS<sub>2</sub> flakes.** (a) Schematic of the ultrafast optical microscopy (UOM) system. (b) A microscope image of single MoS<sub>2</sub> flakes taken with our UOM setup. The white dashed triangle outlines a single MoS<sub>2</sub> flake. (c,d) Transmitted probe images at 550 nm and 650 nm, respectively, with the pump blocked. The scale bar is 10  $\mu\text{m}$ . (e) Photoluminescence spectrum from a monolayer MoS<sub>2</sub> flake. The A exciton peak (1.85 eV), B exciton peak (2.02 eV), and a peak due to defects (1.57 eV) are marked, and the solid lines show Gaussian fits to the data.

### Experimental setup and sample description

UOM is based on conventional optical pump-probe spectroscopy, with the incorporation of tightly focusing lenses or objectives to directly monitor samples with high spatial resolution (Fig. 1(a)). In our system, a 100 kHz regeneratively amplified laser producing 50 femtosecond (fs), 10  $\mu\text{J}$  pulses at 800 nm is used to pump a visible optical parametric amplifier (OPA)<sup>21</sup>. The signal wavelength from the OPA was tuned to generate linearly polarized probe pulses (with polarization perpendicular to the pump pulses) from 550 nm (2.24 eV) to 700 nm (1.77 eV), covering the bandgap (both indirect and direct) of MoS<sub>2</sub>, as well as the well-known A and B exciton transitions<sup>16,26</sup>. The MoS<sub>2</sub> flakes were photoexcited at 400 nm (3.1 eV), well above the bandgap, with pump fluences ( $F$ ) ranging from  $\sim 95$ – $500 \mu\text{J}/\text{cm}^2$ . The pump and probe spot diameters were 63  $\mu\text{m}$  and 30  $\mu\text{m}$ , respectively, which fully covered a single MoS<sub>2</sub> flake without overlapping other flakes. The normalized photoinduced change in transmission ( $\Delta T/T$ ), which is directly proportional to the photoexcited carrier population at a given point in time and space<sup>15,21,27</sup>, is measured at the detector. Real time monitoring of both focused spots and the sample position using a CCD camera with a 50X objective and accompanying beam expander allows us to selectively measure ultrafast carrier dynamics in individual nanomaterials, removing the detrimental influence of inhomogeneous broadening<sup>12,13</sup>.

Images of the atomically thin monolayer (0.6 nm in thickness) MoS<sub>2</sub> flakes taken with our UOM system are shown in Fig. 1(b–d). Briefly, MoO<sub>3</sub> particle-covered SiO<sub>2</sub> substrates were placed in a furnace with a N<sub>2</sub> gas flow, and heated up to 850  $^{\circ}\text{C}$  in the presence of a sulfur source. The as-grown monolayer MoS<sub>2</sub> flakes were then transferred onto MgO substrates for optical measurements via the commonly used polymer supported method (PMMA). An image taken in our system shows the triangular shape of typical monolayer MoS<sub>2</sub> flakes, in which each side is  $\sim 15$ – $20 \mu\text{m}$  in length (Fig. 1(b)). Figure 1(c,d) show images of the transmitted probe beam at 550 nm and 650 nm (with the pump blocked), confirming that the probe beam completely covers single triangular MoS<sub>2</sub> flakes. We also examined double monolayer and multiple layer MoS<sub>2</sub> flakes in our experiments. Double monolayer MoS<sub>2</sub> flakes consist of overlapping triangular monolayer flakes with different orientations and thus appear flower shaped (please see Supplementary Figure S1), and multiple layer flakes consist of overlapped monolayers and bilayers, possibly including trilayers, causing them to appear star or flower shaped. Double and multiple layer flakes have many corners and grain boundaries, allowing us to distinguish between flakes with different layer thicknesses in our UOM system. All experiments were performed at room temperature. Photoluminescence (PL) spectra were also taken on all samples. Figure 1(e) shows the PL spectrum for monolayer MoS<sub>2</sub> at room



**Figure 2. Ultrafast carrier dynamics in single monolayer MoS<sub>2</sub> flakes for different probe photon energies.** Photoinduced changes in transmission for various probe photon energies in a single flake of monolayer MoS<sub>2</sub>. Black dashed lines denote exponential curve fits for various probe photon energies. The curves are displaced vertically for clarity.

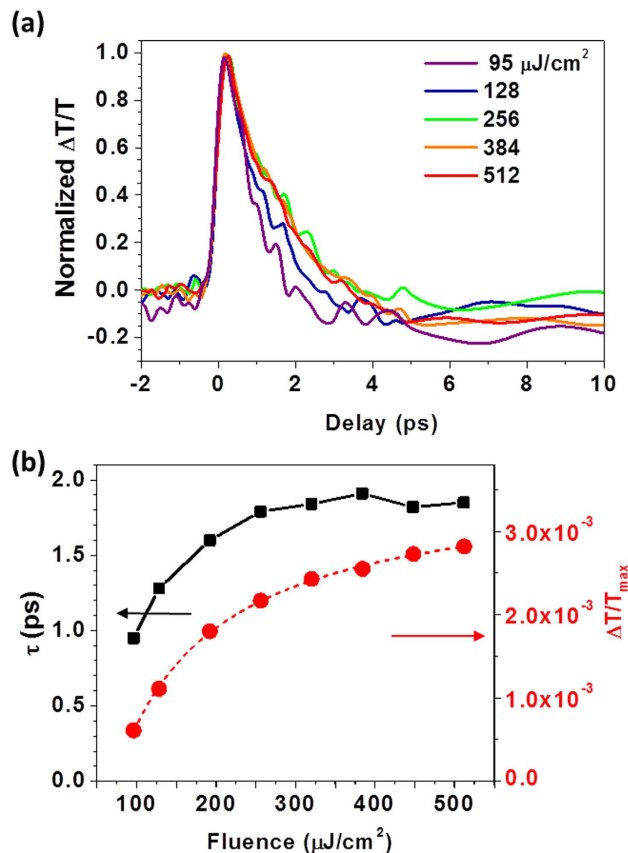
temperature, revealing the A exciton peak at  $\sim 1.85$  eV and the B exciton peak at  $\sim 2.02$  eV. The small lower energy peak at 1.57 eV is due to defects<sup>28</sup>.

## Results

UOM experiments were performed on individual MoS<sub>2</sub> flakes with different thicknesses as a function of the probe photon energy, with the pump fluence initially fixed at  $320 \mu\text{J}/\text{cm}^2$  (producing an initial photoexcited carrier density of  $n \sim 10^{14} \text{ cm}^{-2}$ , calculated using the absorbance in ref. 29). The time-dependent  $\Delta T/T$  signal obtained from a single flake of monolayer MoS<sub>2</sub> for selected photon energies is depicted in Fig. 2. The photoinduced change in transmission reaches a maximum within  $\sim 500$  fs, after which the subsequent fast relaxation occurs on a time scale of a few ps. Photoinduced changes in transmission above and near the bandgap (probe photon energy  $\hbar\omega \geq 1.9$ – $2.25$  eV) on a single flake of monolayer MoS<sub>2</sub> are initially positive, and the  $\Delta T/T$  signal amplitude increases with the probe photon energy. This is expected since there is a greater population of carriers at higher energies than lower energies (i.e., the initial photoexcited carrier population at 3.1 eV loses carriers to non-radiative recombination as carriers relax to lower energies). This positive signal rapidly decays within a few picoseconds (ps), followed by a small negative signal at lower photon energies. When decreasing the probe photon energy to 1.84 eV, near the A excitonic peak ( $\sim 1.85$  eV), we observe a positive peak that decays within a few hundred fs, with a residual negative signal at longer time delays. At even lower photon energies (1.77 eV), the signal is always negative, recovering on a longer time scale of tens of picoseconds.

Physically, the pump pulse initially photoexcites carriers in the MoS<sub>2</sub> monolayer to higher energy states (primarily the C exciton states near the  $\Gamma$  point<sup>14,16,29,30</sup>). The initial photoexcited carrier density is comparable to the Mott density of  $n \sim 10^{14} \text{ cm}^{-2}$ , suggesting that the Coulomb interactions between carriers are largely screened and the photoexcited electrons and holes are no longer bound into excitonic states<sup>18,31</sup>. The non-equilibrium electrons rapidly scatter towards the conduction band minimum at the K point<sup>14–16</sup>, leading to a positive signal for probe photon energies above the bandgap ( $\hbar\omega \geq 1.9$  eV) through state filling (Fig. 2). In contrast, the photoexcited holes have a relatively low probability of scattering to the K point<sup>14</sup>, since the valence band maximum is comparable in energy at both the  $\Gamma$  and K points (see, e.g., Fig. 1 in ref. 30). Therefore, a significant fraction of the non-equilibrium holes likely remain near the  $\Gamma$  point at early times, leading to the negative peak observed within the first few ps at 1.77 eV (through induced absorption). Finally, the initial positive peak observed at 1.84 eV is likely due to the interplay between bandgap renormalization, which has recently been shown to influence the observed dynamics at these carrier densities<sup>18</sup>, and incompletely screened excitons.

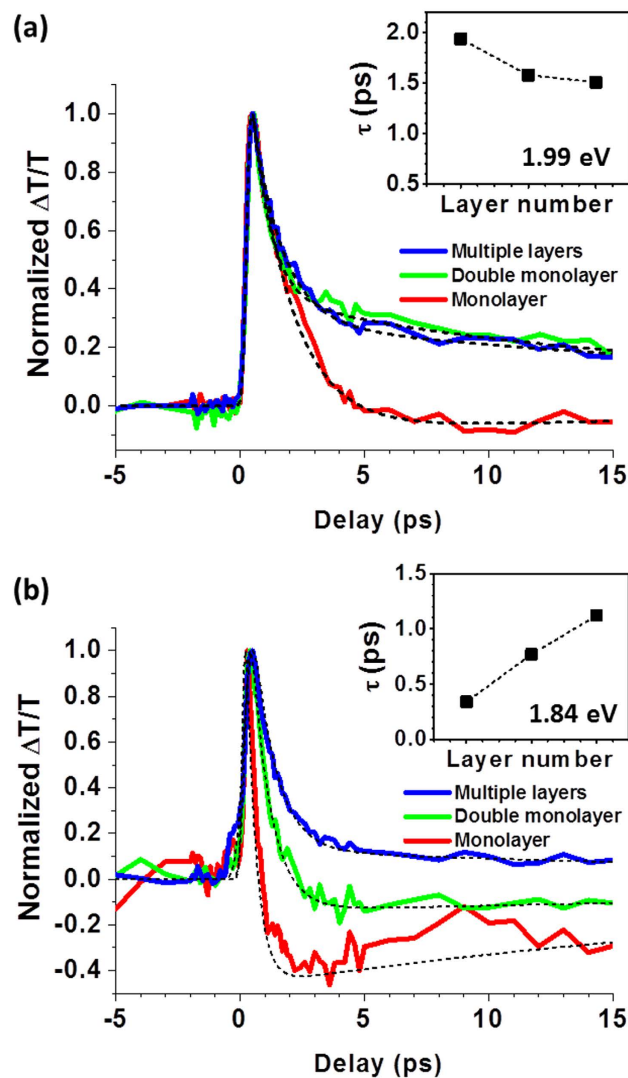
At the highest probe photon energies examined here ( $\hbar\omega \geq 2.14$  eV), the positive  $\Delta T/T$  signal recovers within  $\sim 2$  ps (obtained from exponential curve fits to the data) as the photoexcited electrons relax to lower energies. At lower photon energies ( $1.9 \text{ eV} \leq \hbar\omega \leq 2.05$  eV), we observe a small negative signal after the initial fast relaxation that recovers within 4–5 ps. This is likely due to induced absorption of holes near the  $\Gamma$  point; ref. 32 demonstrated that after photoexciting a MoS<sub>2</sub> monolayer at 3.1 eV, photoluminescence at the A exciton peak appears and reaches a maximum within  $\sim 5$  ps (limited by their experimental resolution). This indicates that photoexcited holes scatter from the  $\Gamma$  point to the K point within this timescale, agreeing well with our data. Finally, the negative signal observed on longer timescales for  $\hbar\omega \leq 1.84$  eV can be attributed to induced absorption from electrons that have relaxed into surface-related defect or trap states<sup>16,23,33</sup>, which should have an appreciable density based on the magnitude of the 1.57 eV peak in our PL data (Fig. 1(e)) (further supported by the fluence-dependent data discussed below). Our curve fits indicate that these carriers relax within  $\sim 30$  ps.



**Figure 3. Pump fluence dependence of ultrafast carrier relaxation in MoS<sub>2</sub> monolayers.** (a) Pump fluence dependence of the photoinduced changes in transmission near the bandgap, 1.91 eV, for a MoS<sub>2</sub> monolayer. (b) Plots of the decay time constant and maximum differential transmission signal extracted from (a) as a function of the pump fluence. The dashed red line represents the fit to the  $\Delta T/T$  signal that allows us to extract  $F_{sat} = 51.5 \mu\text{J}/\text{cm}^2$ <sup>40–41</sup>.

We also examined the dependence of the observed carrier dynamics on pump fluence for the single MoS<sub>2</sub> monolayer flake, in a regime ( $F \sim 95\text{--}500 \mu\text{J}/\text{cm}^2$ ) that has not been extensively studied<sup>17,18,34,35</sup>. Figure 3(a) shows the normalized transmission changes for various pump fluences at a probe photon energy of  $\hbar\omega = 1.91$  eV, revealing that the initial decay time  $\tau$  increases as a function of pump fluence. This is quantified in Fig. 3(b), which shows that  $\tau$  increases by a factor of two and the maximum value of  $\Delta T/T$  increases by more than a factor of three over the measured fluence range, saturating at the highest fluences (likely due to state filling near the band edge). This dependence of  $\tau$  on  $F$  is the opposite of what one would expect for interband Auger recombination, a three-carrier non-radiative process that often dominates carrier relaxation at high fluence in semiconductor nanostructures<sup>19–23,27</sup>. This is also inconsistent with Auger-induced capture into defect states, as described in refs.<sup>19,27</sup>, which predicts the carrier lifetime to decrease with increasing fluence.

Instead, the observed fluence dependence is likely due to one or both of two processes. Ref. 17 observed a similar increase in the carrier lifetime with fluence in few-layer MoS<sub>2</sub> over a comparable fluence range. This was attributed to the hot phonon effect, in which a large phonon population, created by the relatively high temperature and pump fluence, inhibits carrier relaxation to lower energies (which typically occurs through phonon emission<sup>21</sup>), since the transient phonon temperature becomes comparable to that of the non-equilibrium electron distribution<sup>36,37</sup>. Carrier relaxation into surface-related defect and trap states can also influence the observed fluence dependence. These defect states are known to affect the electronic and optical properties of transition metal dichalcogenide monolayers<sup>28,38</sup>, and the relatively large defect-related PL peak in Fig. 1(e) suggests that they will also influence the properties of our MoS<sub>2</sub> monolayers. For low pump fluences ( $F < 250 \mu\text{J}/\text{cm}^2$ ), photoexcited carriers can quickly relax into these defect states (within  $\tau \sim 1\text{--}1.5$  ps). However, the defect states will be filled at high pump fluences, causing  $\Delta T/T$  to saturate and  $\tau$  to increase (since photoexcited carriers in states near the band edge that are examined by the probe photons have nowhere else to relax to before recombining on longer timescales) (Fig. 3(b))<sup>16,39</sup>. The data in Fig. 3 shows that  $\tau$  does not change significantly after  $F \sim 256 \mu\text{J}/\text{cm}^2$  (corresponding to  $n \sim 8 \times 10^{13} \text{cm}^{-2}$ ), suggesting that the defect states are filled for these higher pump fluences. We can gain more insight by fitting our fluence-dependent  $\Delta T/T$  data (Fig. 3(b)) to obtain the saturation fluence  $F_{sat}$ <sup>40,41</sup>. This gives  $F_{sat} = 51.5 \mu\text{J}/\text{cm}^2$ , corresponding to a saturated carrier density of  $n_{sat} \sim 1.6 \times 10^{13} \text{cm}^{-2}$ . Ref. 38 indicates that the defect density in CVD-grown MoS<sub>2</sub> monolayers is  $\sim 10^{13} \text{cm}^{-2}$ , very close to  $n_{sat}$ . This supports the idea



**Figure 4. Ultrafast carrier dynamics in MoS<sub>2</sub> flakes at different probe photon energies for different layer thicknesses.** (a) Normalized photoinduced changes in transmission at 1.99 eV for different MoS<sub>2</sub> samples. The monolayer sample is identical to that described in Figs 1 and 2. The ‘double monolayer’ sample consists of monolayers in contact with one another, while the multilayer sample consists of multiple layers of MoS<sub>2</sub> grown on top of one another. (b) Normalized photoinduced changes in transmission at 1.84 eV for the different MoS<sub>2</sub> samples. The insets show the dependence of the decay time constant on the layer thickness.

that defect and trap states dominate carrier relaxation in our MoS<sub>2</sub> monolayers, although we cannot rule out contributions from the hot phonon effect. Temperature-dependent measurements<sup>17</sup> and tuning the probe wavelength to the defect state PL peak at 1.57 eV could enable us to gain more insight into this in the future. It is worth noting that we observed similar trends at other probe photon energies (e.g., 1.99 eV and 2.25 eV).

Finally, the well-known variation of the electronic structure in MoS<sub>2</sub> with number of layers (e.g., a shift from an indirect gap in multilayer MoS<sub>2</sub> to a direct band gap in monolayer MoS<sub>2</sub>) makes it worthwhile to investigate carrier relaxation in samples with different layer configurations. Photoinduced changes in transmission were measured in isolated MoS<sub>2</sub> monolayers, double monolayers, and multilayer MoS<sub>2</sub> using various probe photon energies. Figure 4(a) depicts the  $\Delta T/T$  signals using a probe photon energy of 1.99 eV. The signal increases with the number of layers because there are more atoms within the excitation spot to absorb the incoming pump photons. The time constant  $\tau$  decreases with layer thickness (Fig. 4(a) inset), consistent with previous measurements<sup>16</sup>. At 1.84 eV, however, the initial  $\Delta T/T$  signal remains positive for the multilayer MoS<sub>2</sub> sample, even at long time delays ( $> 10$  ps) (Fig. 4(b)), in contrast with the rapid transition to a negative value for the isolated monolayer, as discussed earlier. In addition,  $\tau$  increases with the number of MoS<sub>2</sub> layers (Fig. 4(b), inset). These trends are due to the transition from a direct to an indirect bandgap with increasing layer thickness<sup>4,5</sup>. As described above, in the MoS<sub>2</sub> monolayer, the 1.84 eV probe photons initially probe carrier dynamics near the conduction band edge at the K point, as described above. In contrast, in the multilayer sample the 1.84 eV probe photons examine a phonon-assisted absorption from the valence band maximum at the K point to the conduction band

minimum between the  $\Gamma$  and K points. The separation of the electron and hole populations in momentum space, as well as the reduced influence of the surface as the number of layers increases, both increase  $\tau$  for the multilayer sample. Our UOM results thus can be linked to the changes in the MoS<sub>2</sub> electronic structure with the number of layers.

## Conclusion

In conclusion, we have used ultrafast optical microscopy to measure carrier dynamics in single MoS<sub>2</sub> flakes with femtosecond temporal resolution and micron spatial resolution. By tuning the probe photon energy through the bandgap of the MoS<sub>2</sub> flakes, we can obtain temporally and spectrally resolved measurements of carrier dynamics that establish the timescale for carrier relaxation after pump photoexcitation. Furthermore, we observed a variation in the carrier dynamics with pump fluence, primarily due to relaxation into surface-related defect and trap states that eventually saturate at high fluences. Finally, our measurements on samples with different numbers of MoS<sub>2</sub> layers are consistent with the known changes in the band structure with layer thickness. Overall, our UOM studies on monolayer and few-layer MoS<sub>2</sub> provide important insight into the physics of these interesting 2D systems.

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## Author Contributions

M.S., H.Y., A.D.M. and R.P.P. developed the concept. M.S. performed the ultrafast optical experiments, with assistance from S.B.-T. The samples were fabricated by H.Y., A.D.M., S.N., P.M.A. and J.L. photoluminescence spectra were measured by J.-C.B., M.S. and R.P.P. analyzed and interpreted the data, with assistance from A.J.T. The manuscript was primarily written by M.S. and R.P.P. with input from H.Y., A.D.M. and A.J.T.

## Additional Information

**Supplementary information** accompanies this paper at <http://www.nature.com/srep>

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